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# **An Introduction to Density-Functional Theory for Experimentalists**

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# Partnership between theory and experiment

Simulating chemical processes computationally can help interpret and inform experiment.



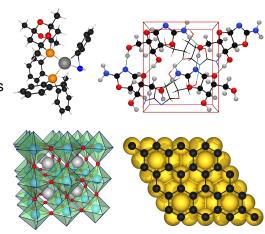




# What can theory predict?

The aim of quantum chemistry is to accurately predict molecular and materials properties from first principles.

- molecular or crystal geometries
- thermodynamic quantities
- mechanisms and chemical kinetics
- spectroscopic quantities
- mechanical properties



## What details do you need to include?

When reading a theory section, you should be able to find enough detail for the calculation to be reproducible.

#### This should include:

- Geometry details for the system under study
- The choice of density functional (and dispersion correction)
- Basis set or plane-wave/pseudopotential details
- Any other specific options

# **Crystals**

Introduction

Geometries of known crystals are available through three databases:



Geometry

Crystallography Open Database

http://crystallography.net/doc



Cambridge Structure Database

http://ccdc.cam.ac.uk



Inorganic Crystal Structure Database

http://icsd.products.fiz-karlsruhe.de

Report the structure code/origin of each cif file.

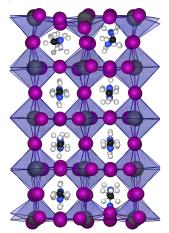
### **Surfaces**

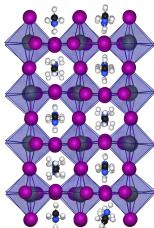
Geometry

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Introduction

Must specify surface termination and any reconstruction – include geometry in the SI.





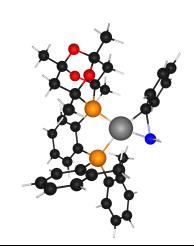
### **Molecules**

Introduction

Optimized xyz coordinates for all structures should appear in the SI.

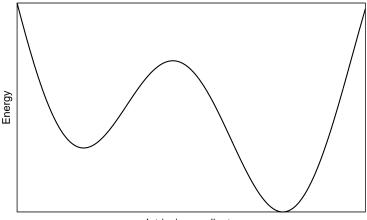
Also include charge/multiplicity when it is not obvious.

```
Ni 0.342223 0.007233 -1.836928
P = 1.222833 = 0.358973 = 0.504922
P 1.955490 0.028431 -0.512822
C -2.187686 1.149487 0.155336
C = 0.249508 = 1.048898 = 0.930789
C = 2.804241 = 1.416203 = 0.614098
C 3.612158 -0.783027 -0.608310
C 1.162462 -0.892194 0.904904
C 2.344338 1.682090 0.208011
C = 3.007420 \ 0.802391 \ 1.399961
0 - 3.098163 1.529544 - 0.903170
C -1.270429 2.338465 0.373163
C -0.824919 -1.777121 1.984884
C = 3.660874 = 0.725084 = 1.686323
0 - 3.567367 - 1.450836 0.616926
```



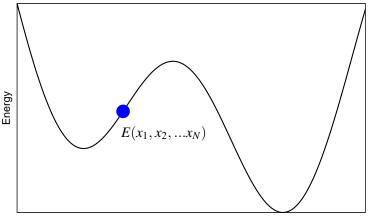
# What types of calculations are possible?

#### Consider a potential energy surface (PES):



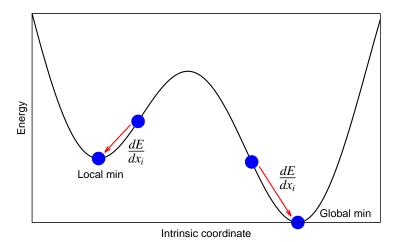
Intrinsic coordinate

# Calculation types — single-point energy

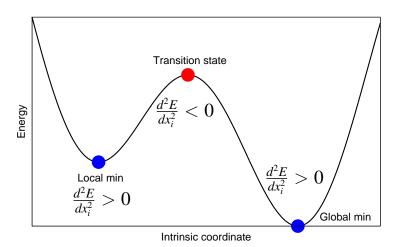


Intrinsic coordinate

# Calculation types — geometry optimization



# Calculation types — frequencies

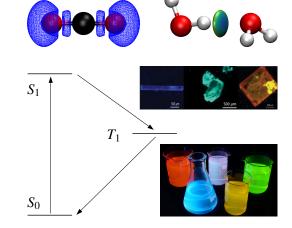


# Other calculation types

Detailed analysis of a particular property can also be done as part of a single-point energy calculation.

Charge analysis

- Energy decomposition
- ELF and NCI plots
- Band structures
- Density of states
- Excitation energies
- NMR shieldings



### **Atomic orbitals**

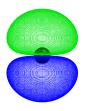
Introduction

Take linear combinations of atomic orbitals to form molecular orbitals:







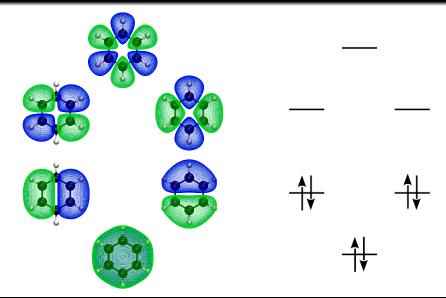


$$MO = c_1 AO_1 + c_2 AO_2 + \ldots + c_n AO_n = \sum_{i=0}^{n} c_i AO_i$$

$$\phi = c_1 \chi_1 + c_2 \chi_2 + \ldots + c_n \chi_n = \sum_{i=0}^n c_i \chi_i$$

Then fill the MOs with electrons.

### **Molecular orbitals**



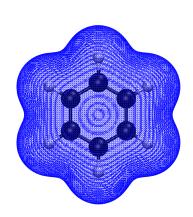
# **Electron density and energy**

The occupied MOs generate the 3D electron density,  $\rho$ .

$$\rho(x, y, z) = \sum_{i} \phi_{i}^{2}(x, y, z)$$

The energy is written as a functional of the density:

$$E = E[\rho(x, y, z)]$$

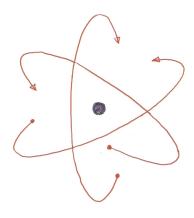


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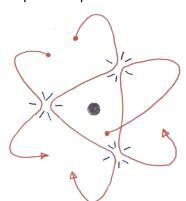
# What is the challenge?

Introduction

Electrons cannot be treated as classical, independent particles.



VS.



# **Density-functional theory (DFT)**

The DFT energy is written as a sum of terms:

$$E = T_0(\rho) + V(\rho) + J_0(\rho) + E_{XC}(\rho)$$

- T<sub>0</sub> is the kinetic energy of the electrons
- V is the electron-nuclear potential energy
- J<sub>0</sub> is the classical electron-electron repulsion energy
- $\bullet$   $E_{\rm XC}$  is the exchange-correlation energy

Exc is the difference between the classical and quantum-mechanical electron-electron interactions.

# **Density-functional approximations (DFAs)**

The DFT energy is written as a sum of terms:

$$E = T_0(\rho) + V(\rho) + J_0(\rho) + E_{XC}(\rho)$$







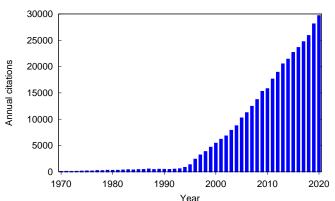


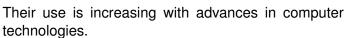
The forms of  $T_0$ , V, and  $J_0$  are known and straightforward to compute.

The exact form of  $E_{\rm XC}$  is unknown and there is no systematic route to obtain it – very many DFAs have been proposed over 4 decades.

# **Density-functional approximations (DFAs)**

While DFT was formulated in 1965, the most popular DFAs were developed between 1986-1996.







Walter Kohn



Axel Becke

# **Hierarchy of DFAs**

Introduction

In order of roughly increasing accuracy and computational time:

- The local spin-density approximation (LSDA)  $E(\rho)$
- Generalized gradient approximations (GGAs)  $E(\rho, \nabla \rho)$ 
  - common examples: BLYP, PW91, PBE, PBEsol, B86bPBE
- meta-GGAs  $E(\rho, \nabla \rho, \nabla^2 \rho, \tau)$ 
  - common examples: BR, M06-L, SCAN
- Global hybrids global mixing of DFAs with Hartree-Fock (HF)
  - common examples: B3LYP, PBE0, M06-2X
- Range-separated hybrids variable mixing of DFAs with HF
  - ► common examples: LC-BLYP, LC-ωPBE, ωB97X

### **Density-functional thermochemistry**

Mean absolute errors, in kcal/mol, for 222 heats of formation:

Method type	MAE
Hartree-Fock	225
LSDA	120
(meta-)GGAs	$\sim$ 10
Hybrids	$\sim$ 2-5

Hybrid functionals are quite accurate for intramolecular bonding.

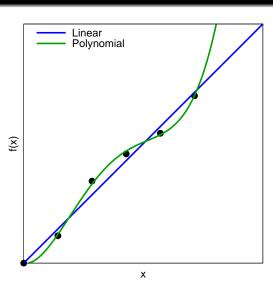
### Physics vs. parameterization

Two schools of thought in functional development:

Introduction

- Physical constraints
- Empirical parameters

"With four parameters I can fit an elephant, and with five I can make him wiggle his trunk."



John von Neumann

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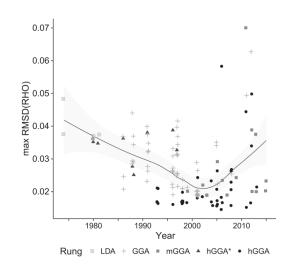
# Physics vs. parameterization

Two schools of thought in functional development:

Introduction

- Physical constraints
- **Empirical parameters**

Too many parameters can lead to over-fitting and errors outside the fit set.



Medvedev et al. Science 355, 49-52 (2017)

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# What type of DFA should I use when?

Solid-state: use GGAs, like PBE or B86bPBE, since hybrids are too expensive.

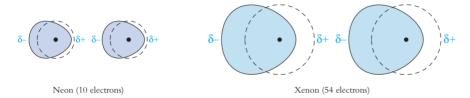
Molecular: use global hybrids like B3LYP or PBE0, unless you have:

- Delocalization/self-interaction error:
  - Examples include charge-transfer complexes, halogen bonding, H-atom transfer transition states, etc.
  - ▶ use range-separated hybrids like LC-ωPBE, ωB97X
- Multi-reference systems/strong correlation error:
  - ► Examples include open-shell singlet biradicals (¹O₂), some transition-metal compounds (Cr<sub>2</sub>)
  - ▶ Use correlated-wavefunction theory, not DFT, if possible
  - Otherwise, use GGAs, like PBE or B86bPBE

# **Dispersion corrections**

Introduction

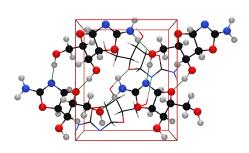
London dispersion is responsible for condensation of non-polar species, like noble gases and hydrocarbons.

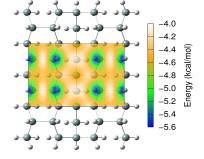


The strength of dispersion interactions increases with polarizability, explaining boiling-point trends.

This long-range non-local interaction is not captured by most density functionals.

# **Dispersion interactions**



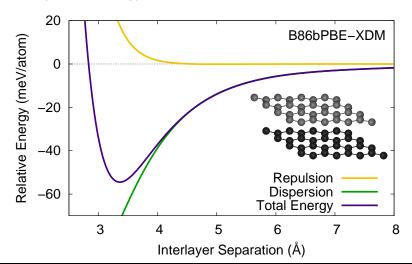


- Biomolecular structure
- Self-assembly
- Layered materials

- Surface adsorption
- Phase transitions
- Crystal packing

# **Dispersion interactions: graphite**

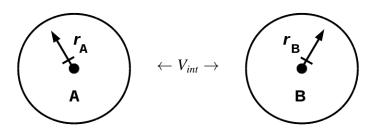
Explicit dispersion-energy terms need to be added to DFT methods.



# The dispersion energy

Introduction

Dispersion arises from interaction of instantaneous dipoles (and higher-order multipoles) in the electron density distribution.



The dispersion energy can be written as a sum over all atom pairs:

$$E_{\text{disp}} = -\sum_{i < j} \left( \frac{C_6}{R_{ij}^6} + \frac{C_8}{R_{ij}^8} + \dots \right)$$

# **Hierarchy of dispersion corrections**

In order of roughly increasing accuracy and computational time:

- D2, TS  $C_6$ 's are fixed or weakly dependent on environment
- D3, D3(BJ) dispersion coefficients depend on coordination
- XDM, MBD highly dependent on environment
- vdW-DF, rVV10 explicitly non-local XC functional

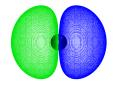
Choosing any is better than nothing — dispersion corrections are physically important and should not be viewed as optional.

### **Atom-centered basis sets**

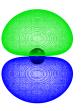
Take linear combinations of atomic orbitals to form molecular orbitals:



Introduction







$$\phi = c_1 \chi_1 + c_2 \chi_2 + \ldots + c_n \chi_n = \sum_{i=0}^n c_i \chi_i$$

However, using hydrogen-like AOs is not practical for computing the electron-electron repulsion integrals.

#### Gaussian basis sets

Use a sum of Gaussians to represent each AO:

$$\chi = Nx^{i}y^{j}z^{k}e^{-ar^{2}}$$

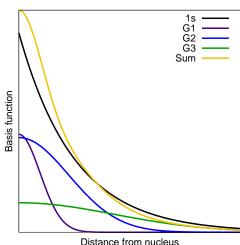
$$i = j = k = 0 \quad \rightarrow \quad \text{s orbital}$$

$$i + j + k = 1 \quad \rightarrow \quad p \text{ orbital}$$

$$i + j + k = 2 \quad \rightarrow \quad d \text{ orbital}$$

$$i + j + k = 3 \quad \rightarrow \quad f \text{ orbital}$$

Using 2 or more sets of Gaussians per orbital allows them to expand or contract.



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# Gaussian basis sets — building flexibility

#### Polarization functions (\*):

- include higher angular-momentum functions
- add p functions to s-block and d functions to p-block elements
- allows polarization of the density along bonds and for lone pairs



Spherical density



Polarized density



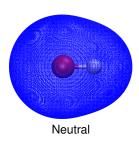
Density difference

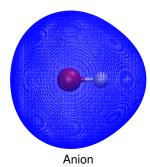


# Gaussian basis sets — building flexibility

#### Diffuse functions (+):

- add broad Gaussians with small exponents
- important for anions, which have large atomic radii
- important for intermolecular interactions, like H-bonding





### Which basis sets should I use?

#### Geometry optimization and frequency calculations:

- time-consuming, require many energy evaluations
- fairly insensitive to basis set
- use small basis sets: 6-31G\*, cc-pVDZ, def-SVP
- for anions and very electronegative atoms, add diffuse functions: 6-31+G\*, aug-cc-pVDZ, def-SVPD

#### Single-point energy calculations:

- fairly quick, require only one energy evaluation
- quite sensitive to basis set
- use large basis sets: 6-311+G(2d,2p), aug-cc-pVTZ, def-TVZPD

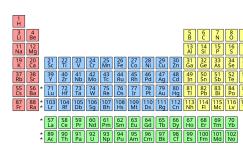
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# **Effective core potentials (ECPs)**

For heavy elements, core electrons are replaced by an ECP to

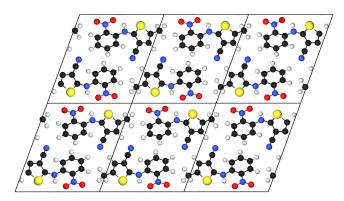
- save computational time since core electrons are unreactive
- include scalar relativistic effects.
- ECPs are typically necessary for  $Z \ge 37$  (i.e. beyond Kr)
- For transition metals, the outermost s and p electrons should be modeled explicitly, not included in the ECP





# **Periodic boundary conditions**

Most solids are crystals than can be represented by a single unit cell, repeatedly replicated in 3D.

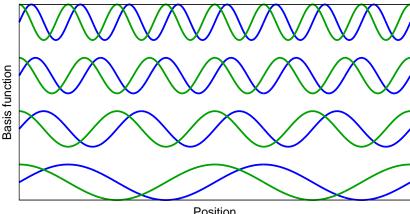


The basis functions should have the same periodicity as the lattice.

### Plane-wave basis sets

Introduction

All plane waves are included up to some energy (frequency) cutoff.



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# **Pseudopotentials**

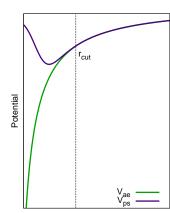
Introduction

Sharp electron-density peaks near nuclei are hard to represent using plane waves.

Replace the potential (and the density) within some cut-off radius,  $r_{\rm cut}$ .

Pseudopotentials are used for all elements.

Core regions between bonded atoms must never overlap.

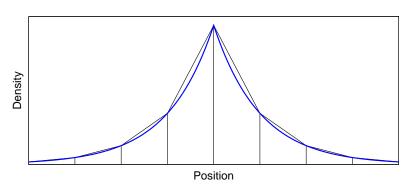


Distance from nucleus

# Integrals: k-point mesh

Introduction

Integrals to evaluate the energy are done numerically – need a dense enough mesh to obtain converged values.



In reciprocal space - more points for small cell dimensions and few points for large cell dimensions.

### Summary

Introduction

DFT is a powerful tool for chemistry, physics, materials science, and engineering, but is not a black box – always seek input from experts.

It is important that enough detail is given for the calculations to be reproducible and for the reader to judge their quality.

- Geometry details for the system under study
  - cif files or xyz coordinates
  - charge and magnetization
- The choice of density functional
  - functional name, citation, dispersion correction
- Basis set or plane-wave/pseudopotential details
  - basis set and ECP
  - pseudopotential type, plane-wave cutoffs, k-point mesh